

Today's outline - November 11, 2024



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- Final Project

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- EXAFS Theory

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Reading Assignment: Chapter 8.4

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Reading Assignment: Chapter 8.4

Homework Assignment #06:

Chapter 6: 1,6,7,8,9

due Friday, November 15, 2024



- Final Project
- EXAFS Theory
- XAS Experiments
- Photoemission

Reading Assignment: Chapter 8.4

Homework Assignment #06:

Chapter 6: 1,6,7,8,9

due Friday, November 15, 2024

Please send me your choices for General User proposal and final exam presentation. I need to approve them by the end of the week!

Final projects & presentations



In-class student presentations on research topics

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In-class student presentations on research topics

- Choose a research article which features a synchrotron technique

Final projects & presentations



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- Schedule a 20 minute time on Final Exam Day (Thursday, May 4, 2023, 17:00-19:00)

Final projects & presentations



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Final project - writing a General User Proposal



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Final project - writing a General User Proposal

- Think of a research problem (could be yours) that can be approached using synchrotron radiation techniques



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Final project - writing a General User Proposal

- Think of a research problem (could be yours) that can be approached using synchrotron radiation techniques
- Make proposal and get approval from instructor before starting



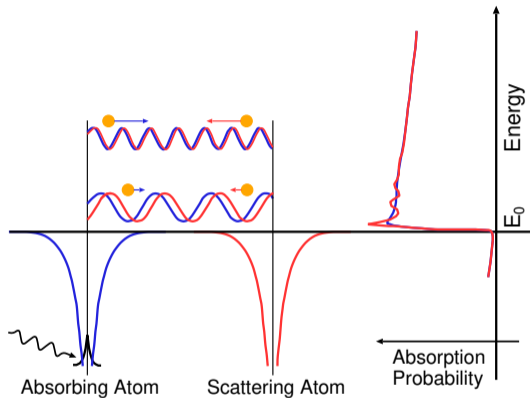
In-class student presentations on research topics

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Final project - writing a General User Proposal

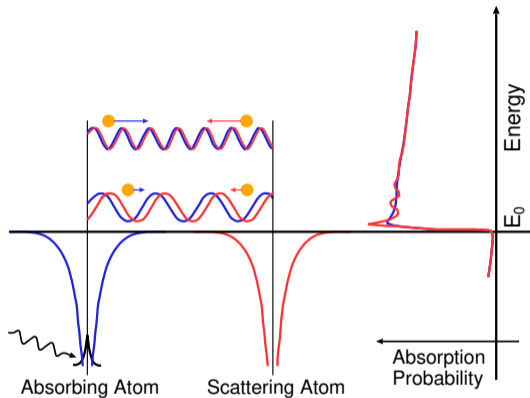
- Think of a research problem (could be yours) that can be approached using synchrotron radiation techniques
- Make proposal and get approval from instructor before starting
- **Must be different technique than your presentation!**

X-ray absorption: Fermi's golden rule



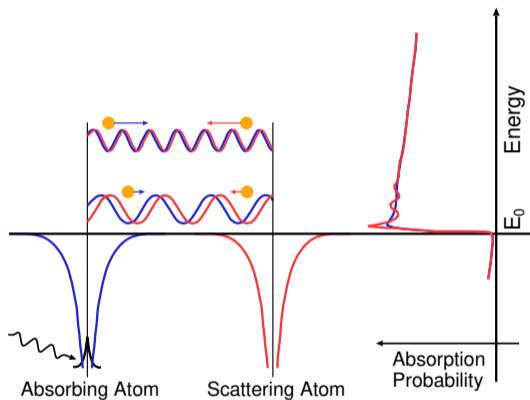
$$\mu(E) = \mu_0(E) + \Delta\mu(E)$$

X-ray absorption: Fermi's golden rule



$$\mu(E) = \mu_0(E) + \Delta\mu(E) = \mu_0(E)[1 + \chi(E)]$$

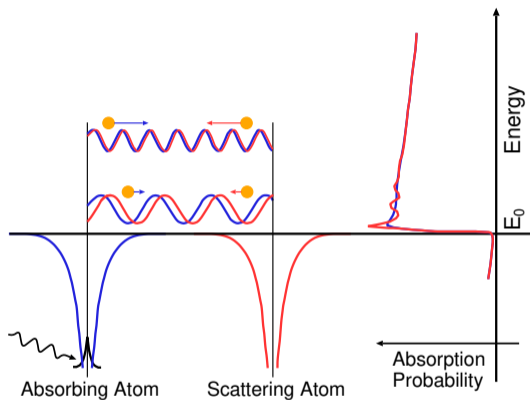
X-ray absorption: Fermi's golden rule



$$\mu(E) = \mu_0(E) + \Delta\mu(E) = \mu_0(E)[1 + \chi(E)]$$

$$\chi(k[E]) = \frac{\mu(E) - \mu_0(E)}{\mu_0(E)},$$

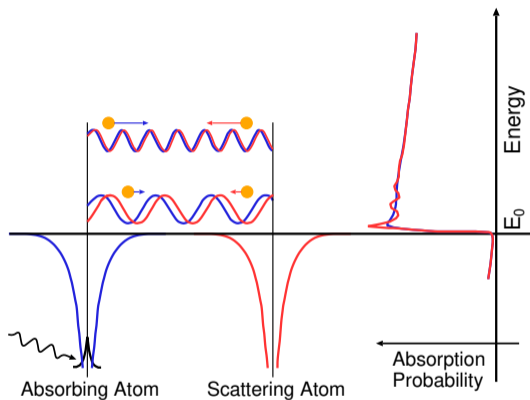
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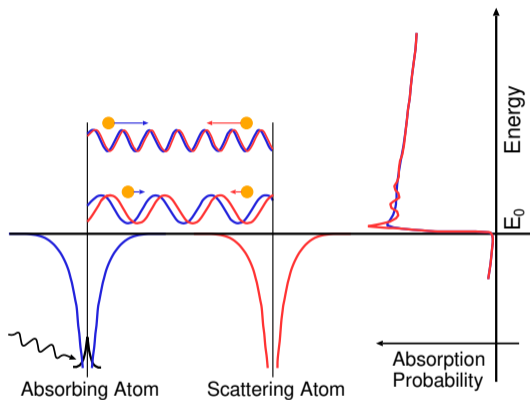


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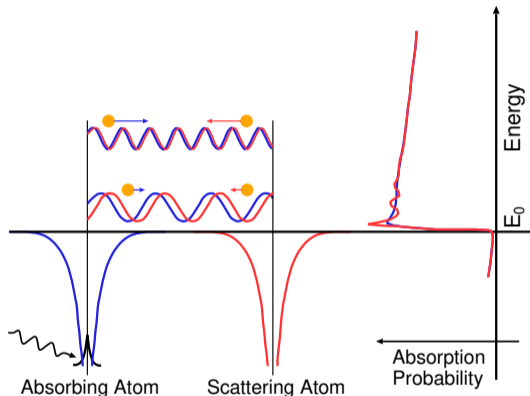
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$\langle i |$ is the **initial state** which has a core level electron and the photon. This is not altered by the neighboring atom.

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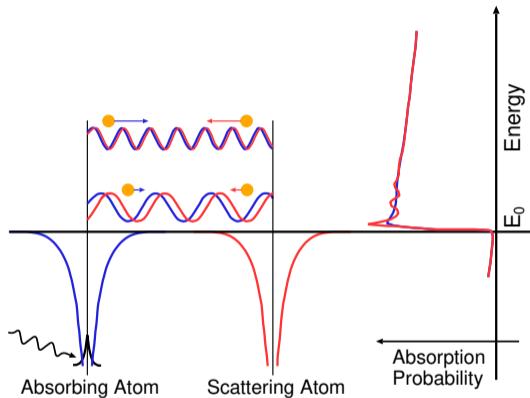
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$|i\rangle$ is the **initial state** which has a core level electron and the photon. This is not altered by the neighboring atom.

\mathcal{H} is the **interaction**. In the dipole approximation, $\mathcal{H} = e^{ikr} \approx 1$.

X-ray absorption: Fermi's golden rule



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\mathcal{H} is the **interaction**. In the dipole approximation, $\mathcal{H} = e^{ikr} \approx 1$.

$|f\rangle$ is the **final state** which has a photoelectron, a hole in the core, and no photon. This is altered by the neighboring atom: the photoelectron scatters.



μ and χ and the photoelectron wavefunction

Writing $|f\rangle = |f_0 + \Delta f\rangle$, where Δf gives the change in photoelectron final state due to backscattering from the neighboring atom, we can expand μ to get

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$$\begin{aligned}\mu_0(E) &\sim |\langle i|\mathcal{H}|f_0\rangle|^2 && \text{atomic background} \\ \chi(E) &\sim \langle i|\mathcal{H}|\Delta f\rangle && \text{XAFS oscillations}\end{aligned}$$



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$$\chi(E) \sim \langle i|\Delta f\rangle \sim \int \psi_{\text{core}} \psi_{\text{scatt}}(r) dr$$



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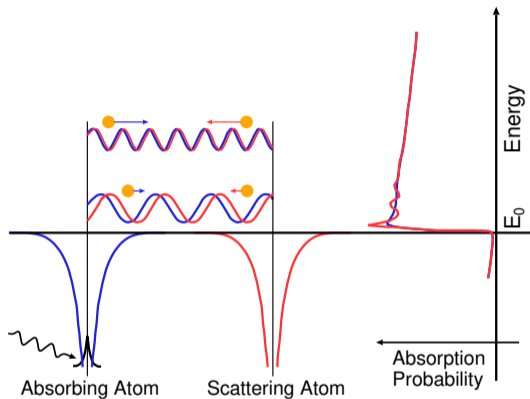
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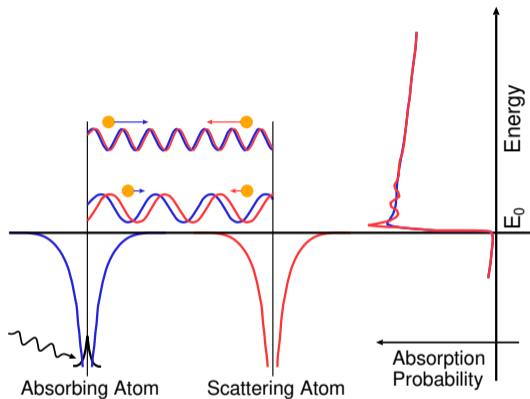
$$\chi(E) \sim \langle i|\Delta f\rangle \sim \int \psi_{\text{core}} \psi_{\text{scatt}}(r) dr \sim \int \delta(r) \psi_{\text{scatt}}(r) dr = \psi_{\text{scatt}}(0)$$

Computing the scattered wavefunction



Assume that emitted photoelectron is a spherical wave

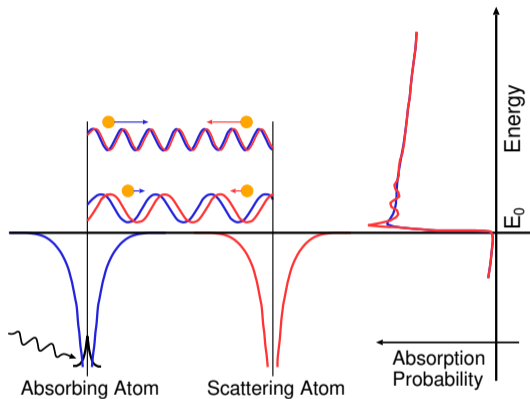
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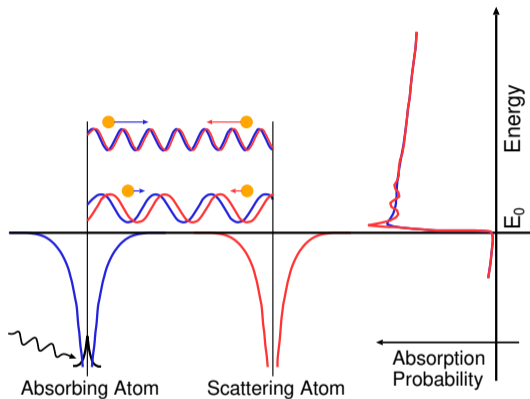


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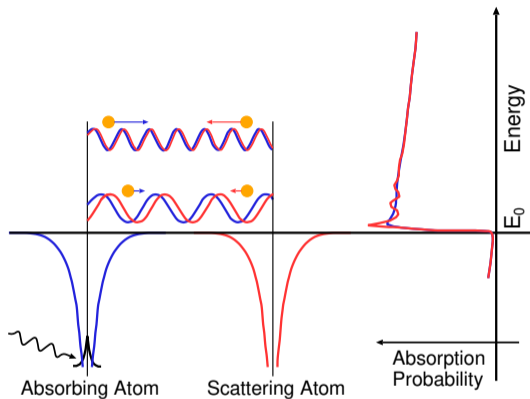
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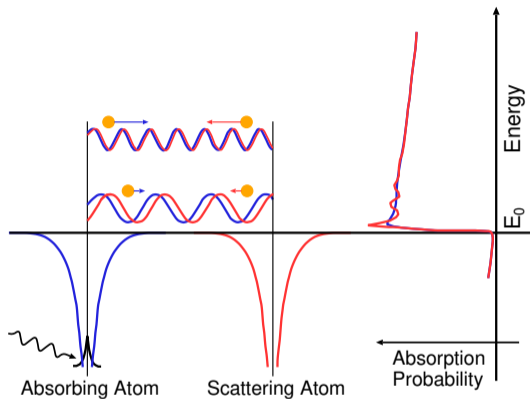
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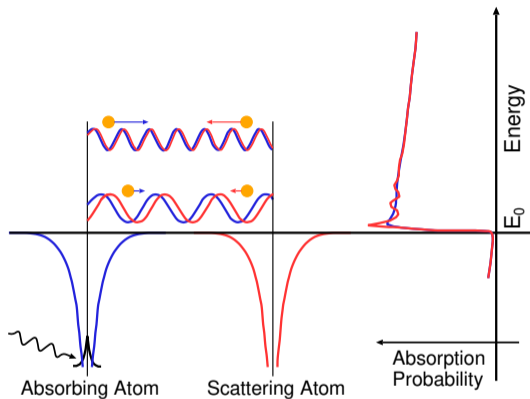
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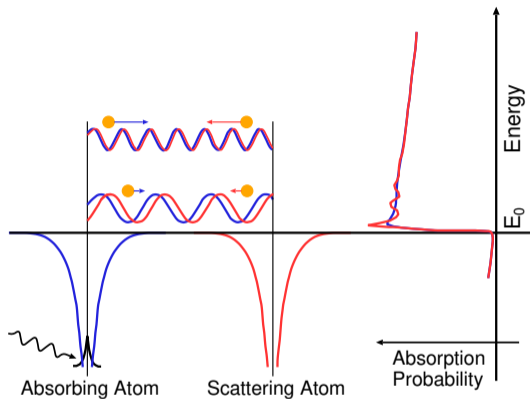
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Development of the EXAFS equation



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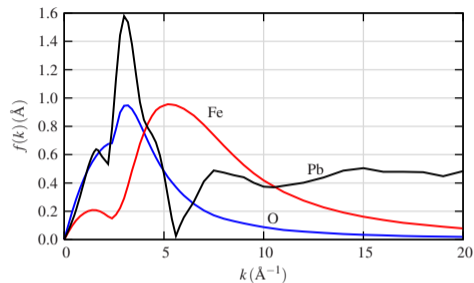


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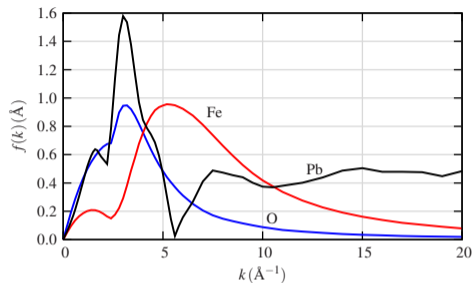


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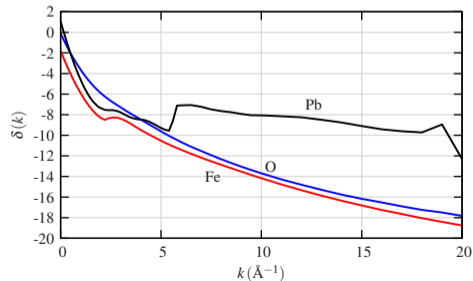
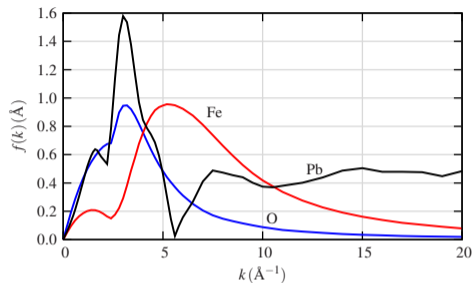


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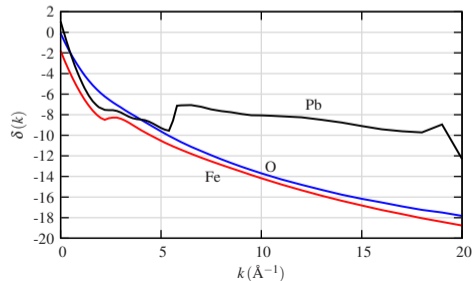
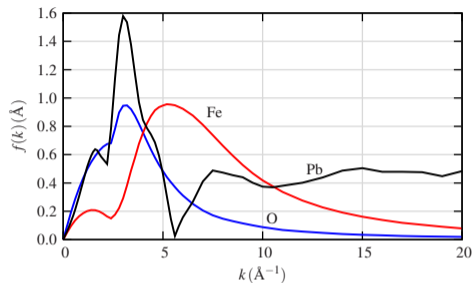


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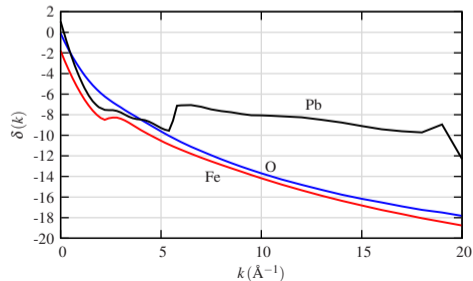
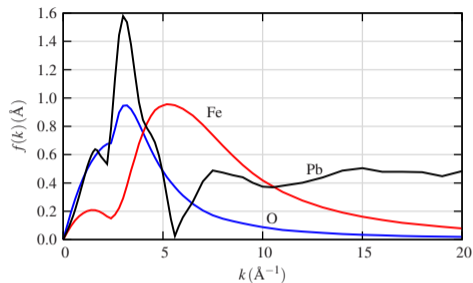
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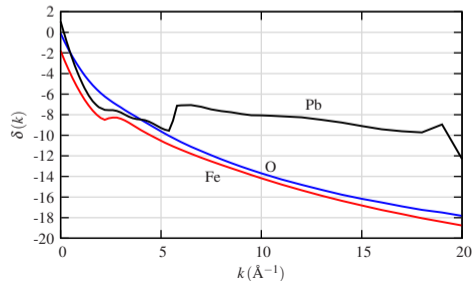
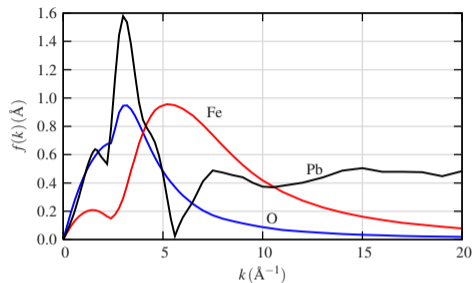
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Z can usually be determined to ± 5 . Fe and O can be distinguished, but Fe and Mn cannot be



The EXAFS equation: What we left out



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Generally, the calculations (FEFF, etc) include these effects. We'll discuss of few of these in more detail . . .

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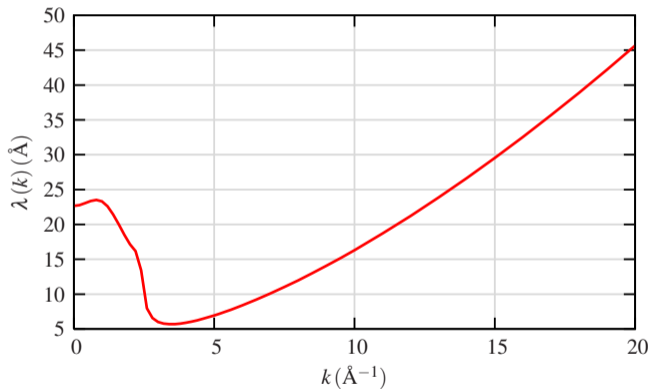
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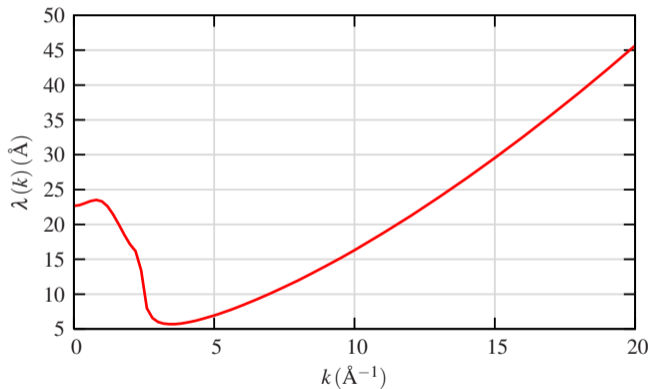
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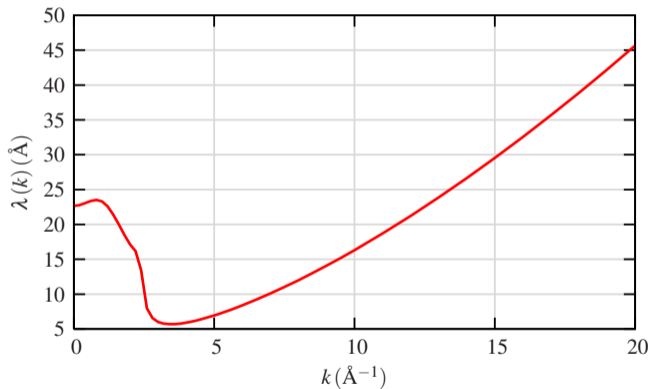
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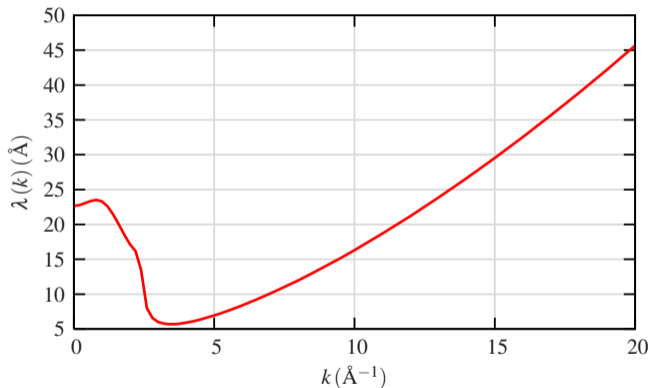


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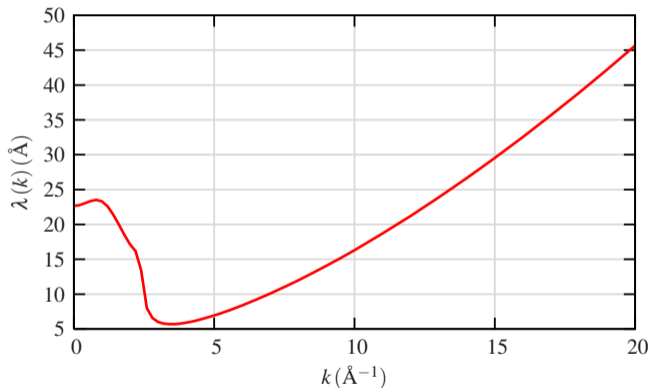
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for XANES ($k < 3 \text{\AA}^{-1}$), both λ
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$\langle \Phi_f^{N-1}| = (N - 1)$ – electrons, relaxed by core-hole

S_0^2 is usually taken as a constant:

$$0.7 < S_0^2 < 1.0$$

and is used as a fitting parameter that multiplies χ :

S_0^2 is completely correlated with N (!!!)

S_0^2 : Amplitude reduction term



Another important **amplitude reduction term** is due to the relaxation of the **other electrons in the absorbing atom** to the hole in the core level:

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This, and other experimental and theoretical issues, make EXAFS amplitudes (and therefore N) less precise than EXAFS phases (and therefore R)

The EXAFS equation



The full EXAFS equation can be used to model and interpret experimental data

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$$\chi(k) = \sum_j \frac{N_j S_0^2 f_j(k) e^{-2R_j/\lambda(k)} e^{-2k^2\sigma_j^2}}{kR_j^2} \sin [2kR_j + \delta_j(k)]$$

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Because we can compute $f(k)$ and $\delta(k)$, and $\lambda(k)$ we can determine Z , R , N , and σ^2 for scattering paths to neighboring atoms by fitting the data.

Sum over paths and multiple scattering



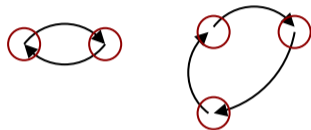
A sum over scattering paths allows **multiple-scattering paths**: the photoelectron scatters from **more than one atom** before returning to the absorbing atom:

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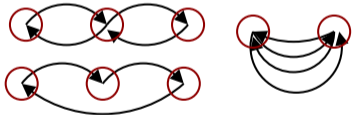


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Single Scattering Triangle Paths



Focused Multiple Scattering Paths

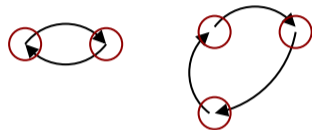


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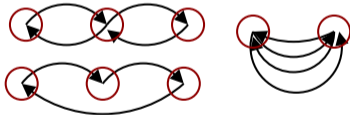
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For multi-bounce paths, the total amplitude depends on the **angles** in the photoelectron path

Focussed Multiple Scattering Paths

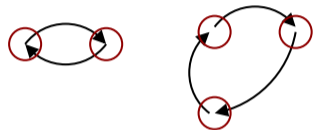


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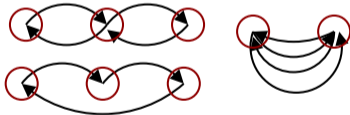
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Triangle Paths with angles $45^\circ < \theta < 135^\circ$ aren't strong, but there can be a lot of them

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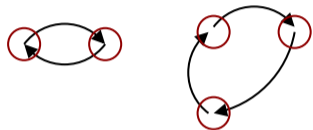


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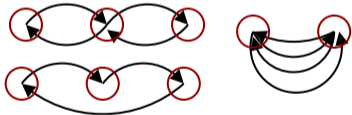
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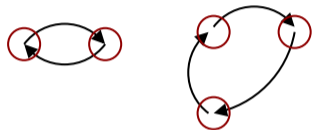
Linear paths, with angles $\theta \approx 180^\circ$, are very strong: the photoelectron can be **focused** through one atom to the next

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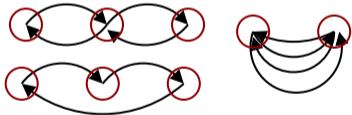
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FEFF calculates these effects and includes them in $f(k)$ and $\delta(k)$ for the EXAFS equation so that all paths look the same in the analysis

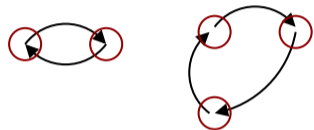
Multiple Scattering is strongest when $\theta > 150^\circ$ and the strong angular dependence can be used to measure bond angles

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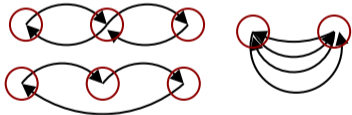
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Multiple Scattering is strongest when $\theta > 150^\circ$ and the strong angular dependence can be used to measure bond angles

For first shell analysis, multiple scattering is hardly ever needed



Sn₄P₃/graphite composite anode

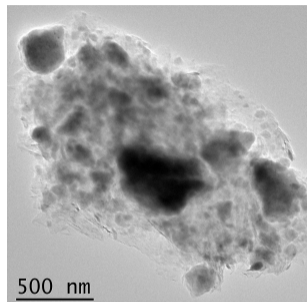
Sn₄P₃ synthesized by high energy ball milling, then ball milled again with graphite to obtain composite

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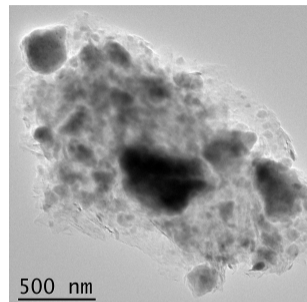
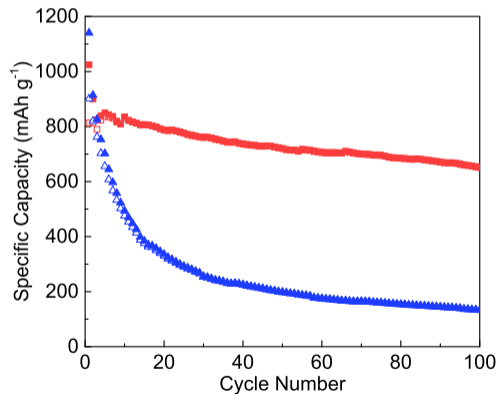


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Sn_4P_3 /graphite composite anode

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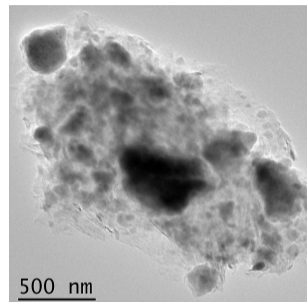
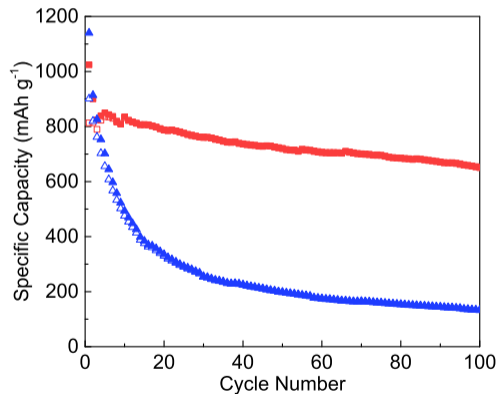


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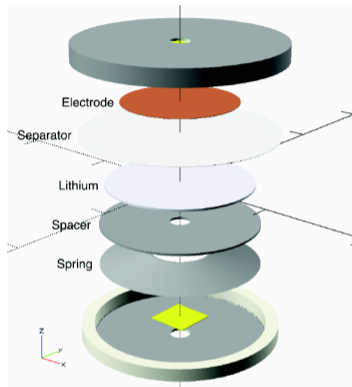
Sn₄P₃ synthesized by high energy ball milling, then ball milled again with graphite to obtain composite



Sn₄P₃/graphite composite shows stable, reversible capacity of 610 mAh/g for 100 cycles at C/2 compared to rapidly fading pure Sn₄P₃ material.

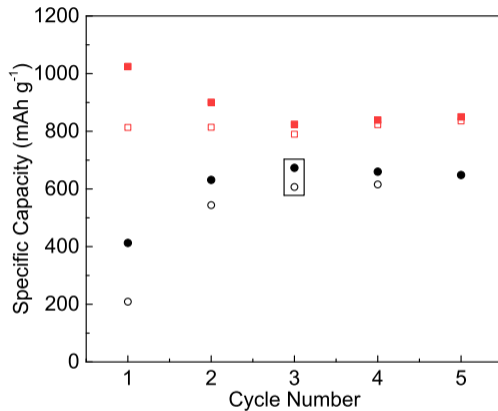
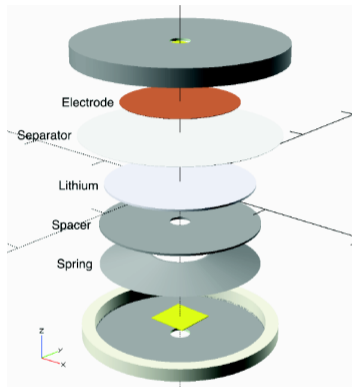
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In situ EXAFS of Sn₄P₃/graphite



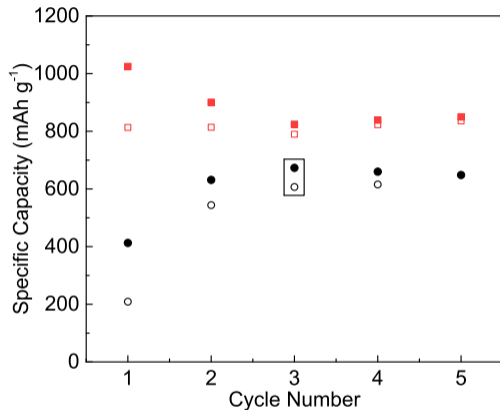
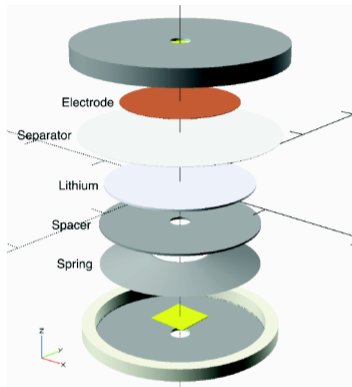
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In situ EXAFS of Sn₄P₃/graphite



Results for *in situ* coin cell are close to the capacity of the unmodified cell at C/4, indicating good reversibility by the 3rd cycle.

"In situ EXAFS-derived mechanism of highly reversible tin phosphide/graphite composite anode for Li-ion batteries," Y. Ding, Z. Li, E.V. Timofeeva, and C.U. Segre, *Adv. Energy Mater.* **8**, 1702134 (2018).

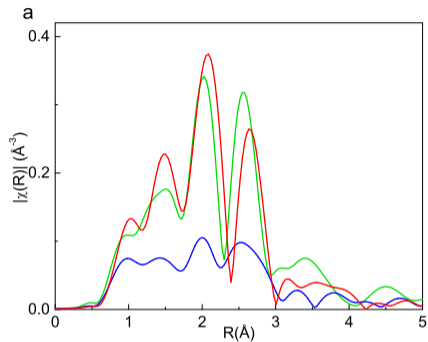
Third cycle comparison



By the third lithiation and third delithiation, the difference between pure Sn_4P_3 and the Sn_4P_3 /graphite composite is clear.

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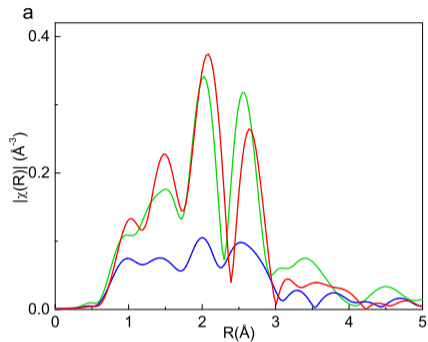


$\text{Sn}_4\text{P}_3/\text{graphite}$ composite

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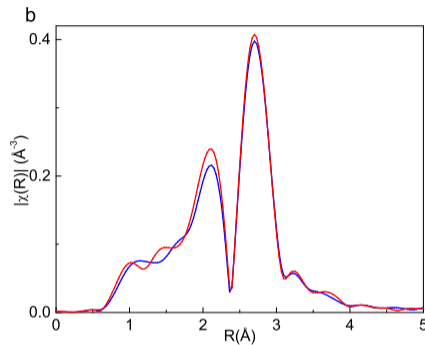
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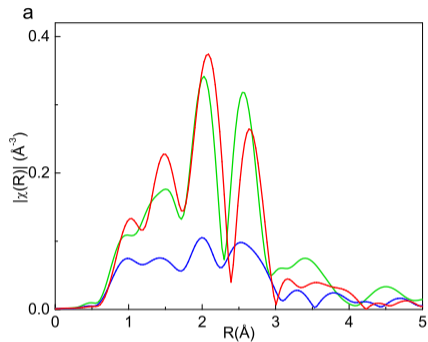
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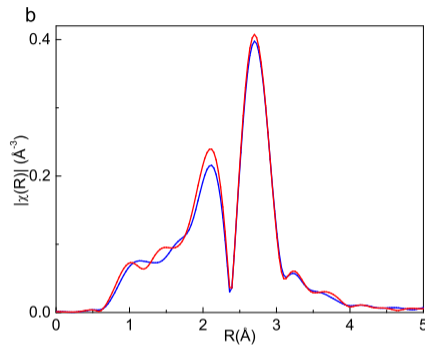
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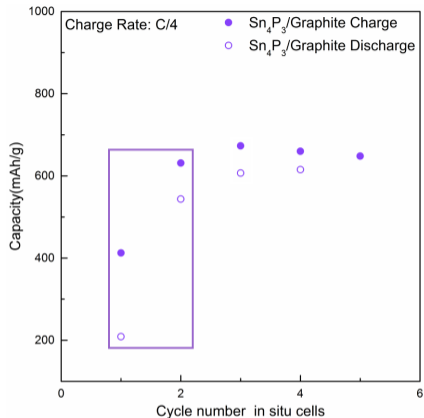
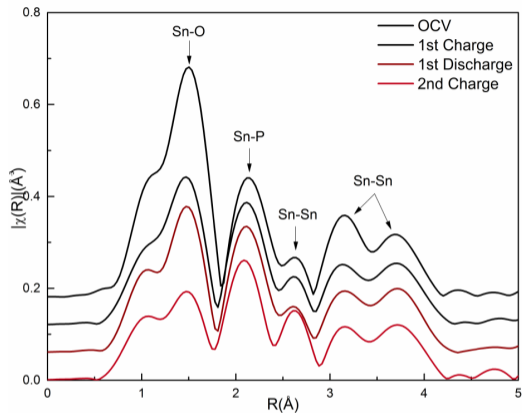


pure Sn₄P₃

Even at the 100th delithiation, the Sn₄P₃/graphite composite measured *ex situ* is showing the same features as at the 3rd cycle.

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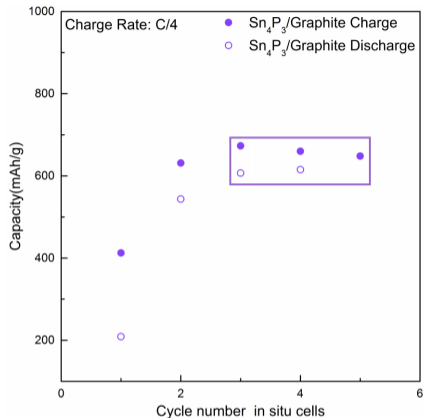
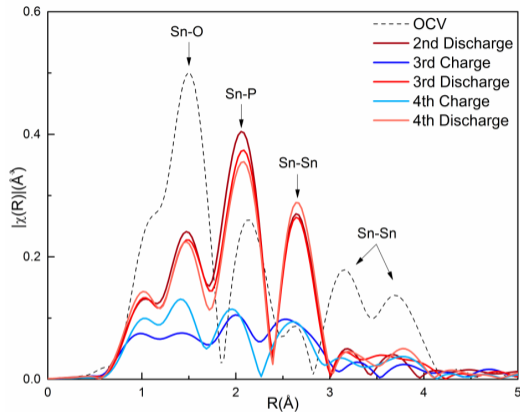
Sn₄P₃/graphite initial cycling



OCV spectrum fits well to Sn₄P₃ structure with an additional Sn-O path

Sn₄P₃ structure persists through first two cycles with possible enhancement of the Sn-Sn path at 2.6 \AA

Sn₄P₃/graphite reversible cycling



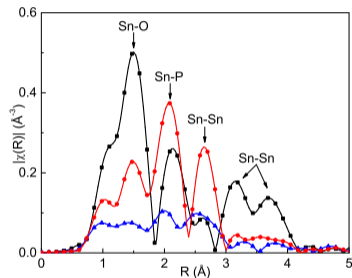
On third lithiation (charge) the Sn-P path is gone and only Sn-Li remains

Delithiation (discharge) produces Sn-P and Sn-Sn paths which are not those of Sn₄P₃ but are reversible

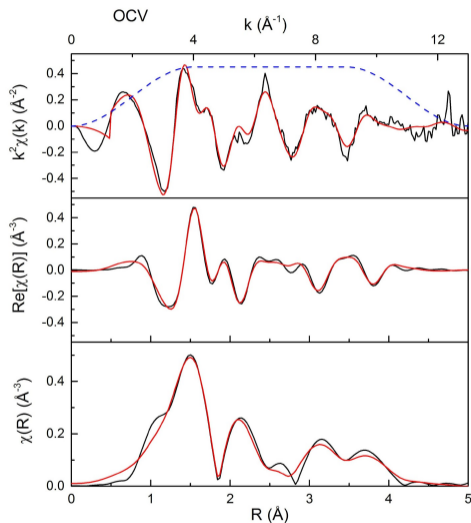
Example fits



Fit EXAFS for bond lengths and coordination numbers



The Sn-O peak at OCV is due to ball milling, which introduces oxygen.

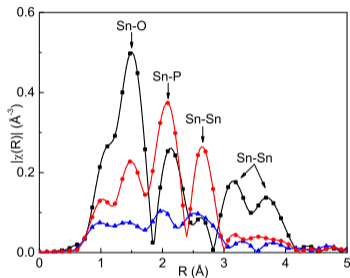


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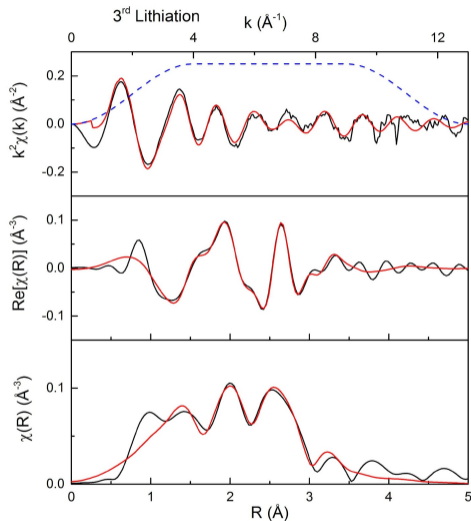


Example fits

Fit EXAFS for bond lengths and coordination numbers



By the 3rd lithiated state, the EXAFS is dominated by Sn-Li paths at 2.7 \AA and 3.0 \AA .

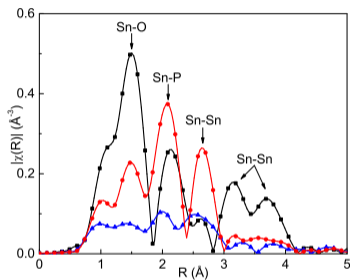


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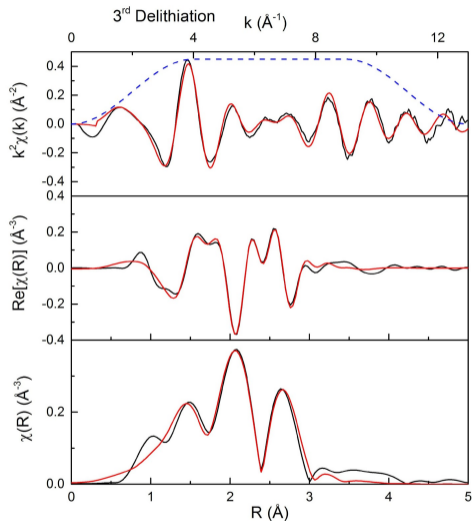


Example fits

Fit EXAFS for bond lengths and coordination numbers

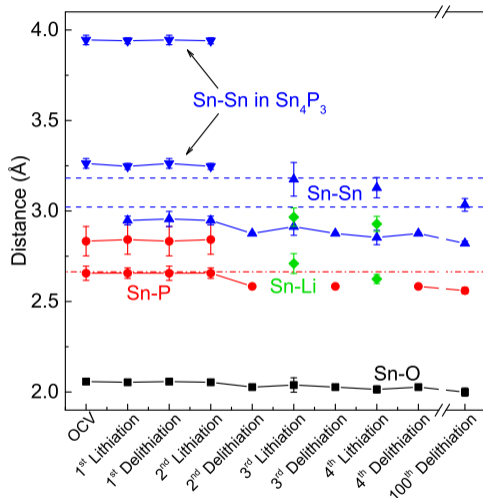


At the **3rd delithiation**, the Sn-P path reappears but at a shorter distance, in an amorphous SnP_x phase.



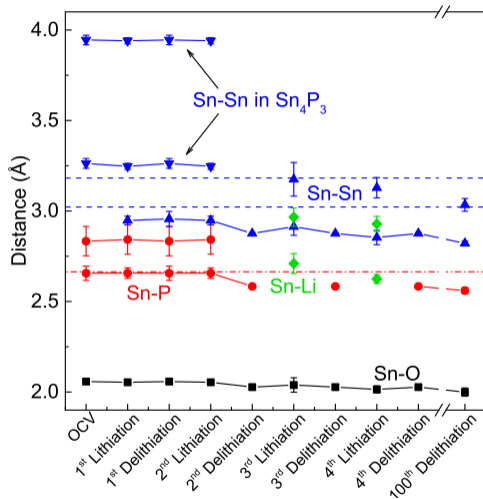
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Sn₄P₃/graphite path lengths



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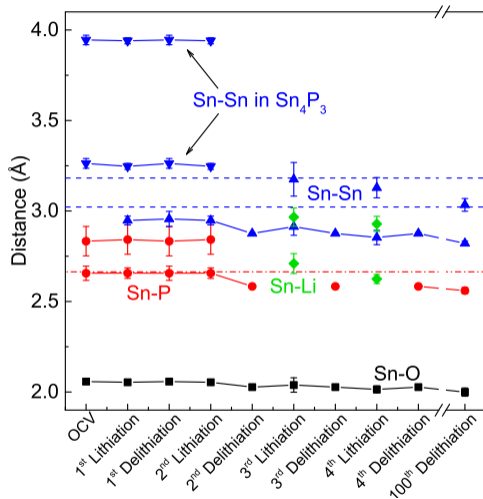
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Sn-Sn distance close to those of metallic Sn indicate the presence of small Sn clusters which may never fully lithiate

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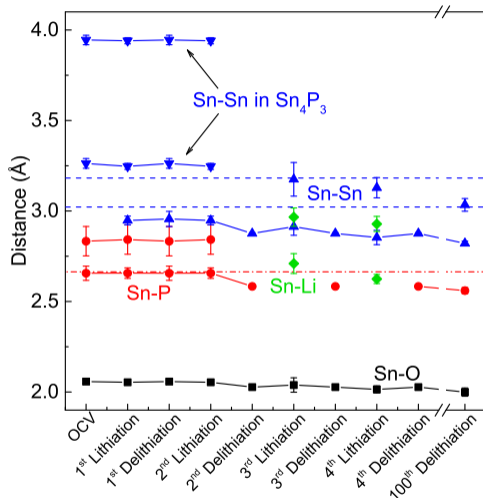


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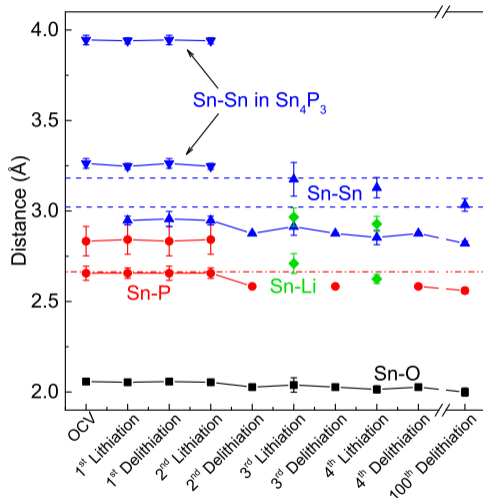
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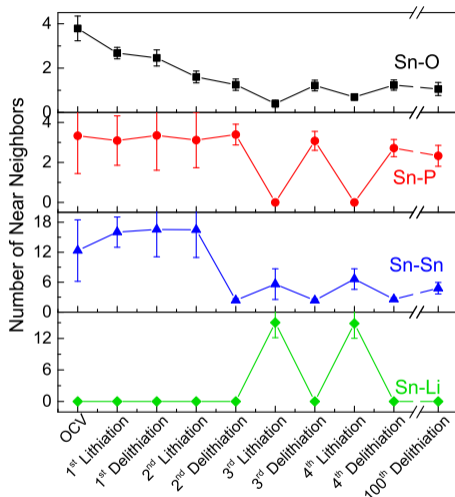
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Only 2 Sn-Li paths present in this material

Sn-O distances remain constant, likely indicative of surface contamination

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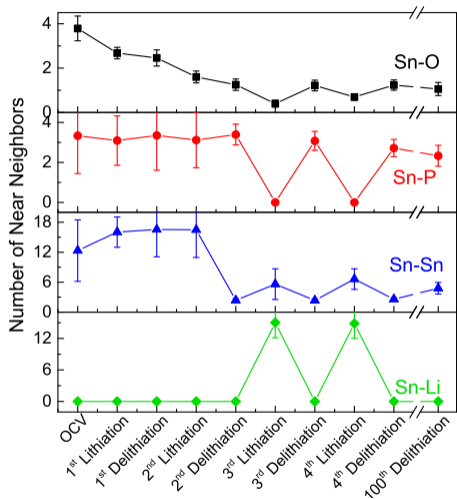
Sn₄P₃/graphite coordination numbers



Sn-O neighbors decrease quickly, remaining small and partially reversible up to 100 cycles

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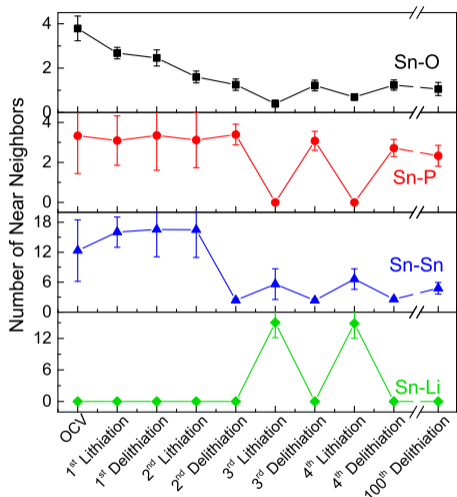


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Sn-P reversible after initial conversion with a slow decrease which correlates to capacity loss

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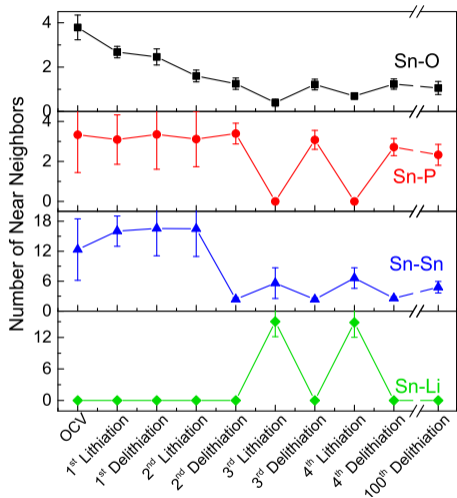
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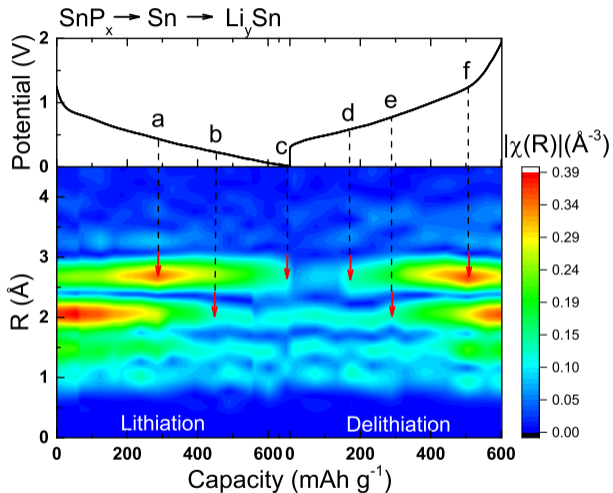
Sn-P reversible after initial conversion with a slow decrease which correlates to capacity loss

Very small Sn-Sn metallic clusters present throughout

The ~ 3.3 Sn-P neighbors in the delithiated state indicate a possibly tetrahedral Sn coordination in SnP_x

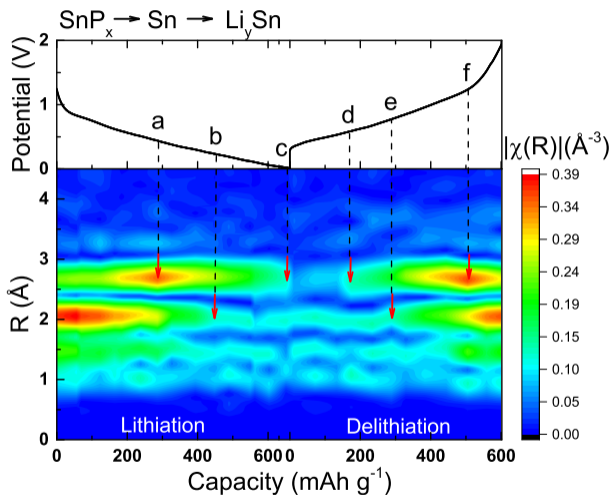
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Third cycle dynamic snapshot

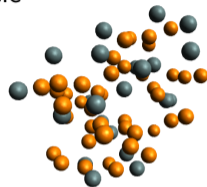


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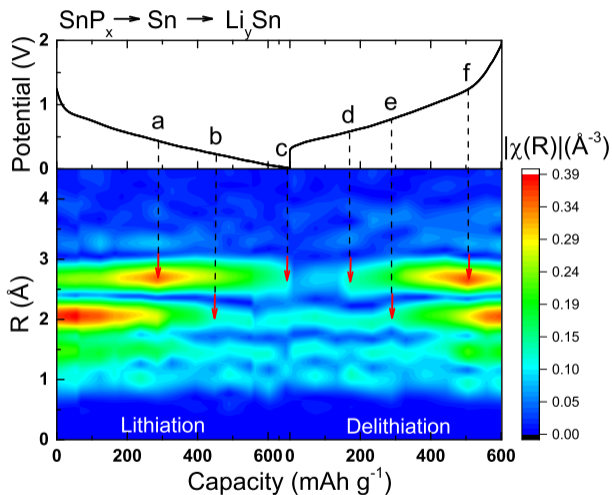


Amorphous SnP_x fully formed at start of 3rd cycle

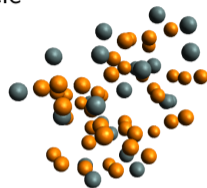


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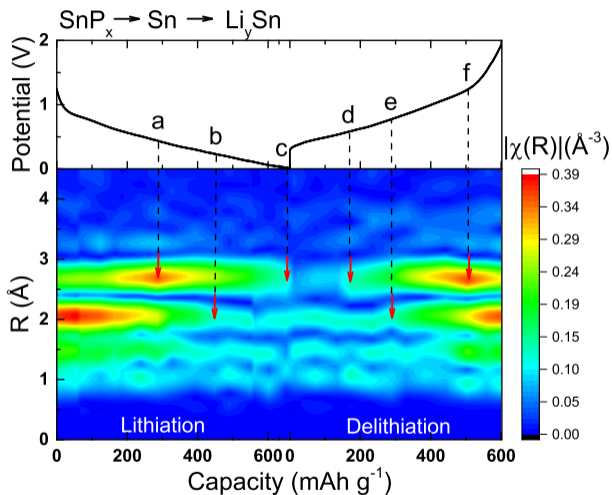
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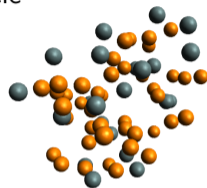
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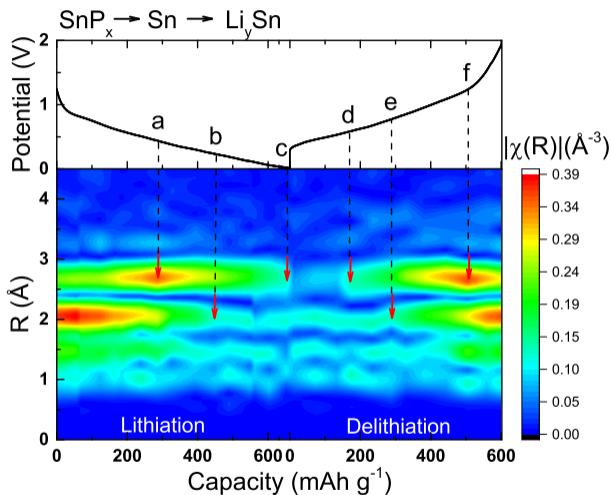
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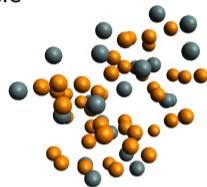
- a. Sn lithiating
- b. SnP_x all gone

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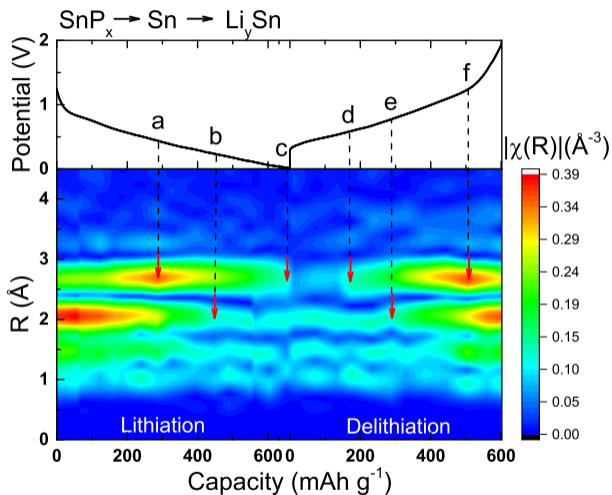
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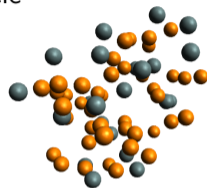
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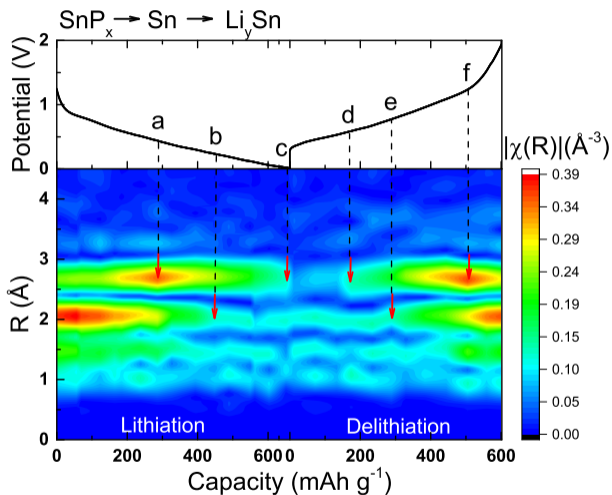
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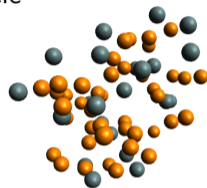
- a. Sn lithiating
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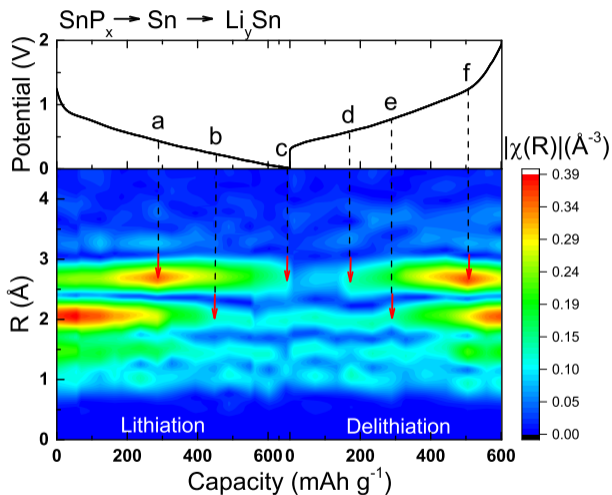
Amorphous SnP_x fully formed at start of 3rd cycle



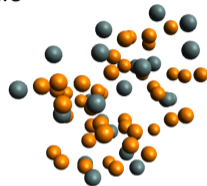
- a. Sn lithiating
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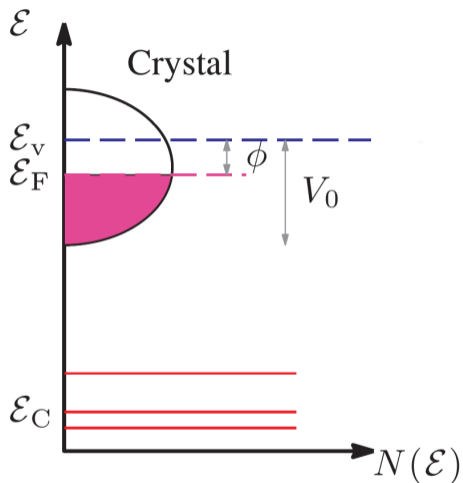
Amorphous SnP_x fully formed at start of 3rd cycle



- a. Sn lithiating
- b. SnP_x all gone
- c. full lithiation
- d. Sn appears
- e. SnP_x appears
- f. Sn delithiated

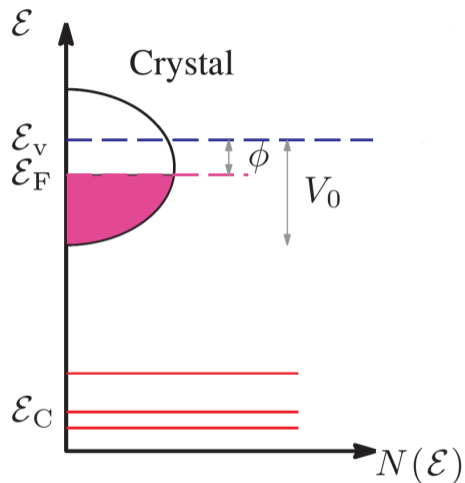
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The photoemission process



Photoemission is the complement to XAFS. It probes the filled states below the Fermi level

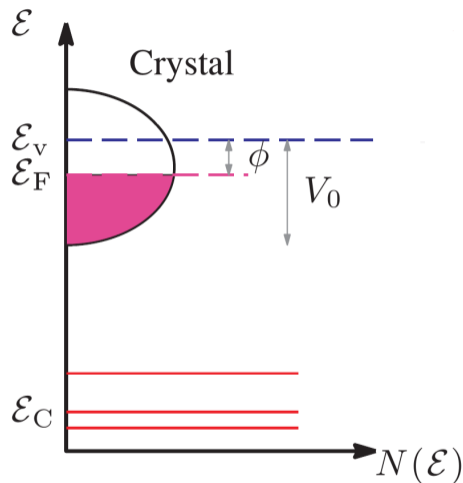
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The dispersion relation of electrons in a solid, $\mathcal{E}(\vec{q})$ can be probed by angle resolved photoemission since both the kinetic energy, \mathcal{E}_{kin} , and the angle, θ are measured

The photoemission process

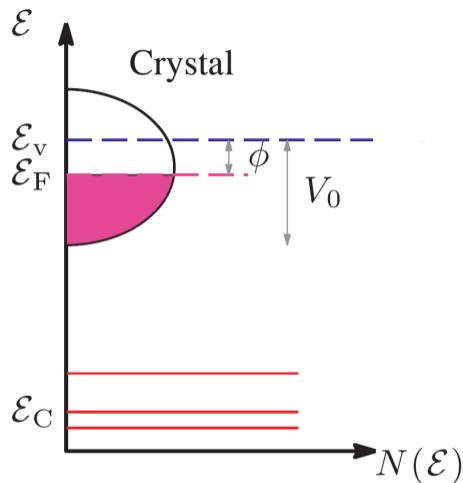


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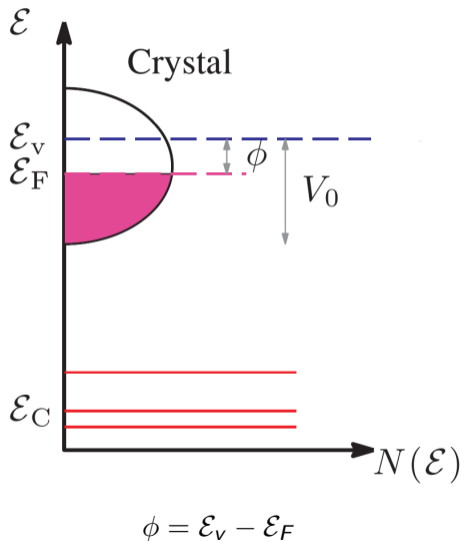
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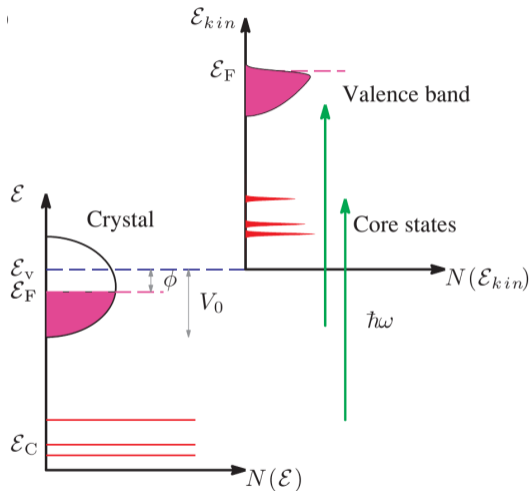
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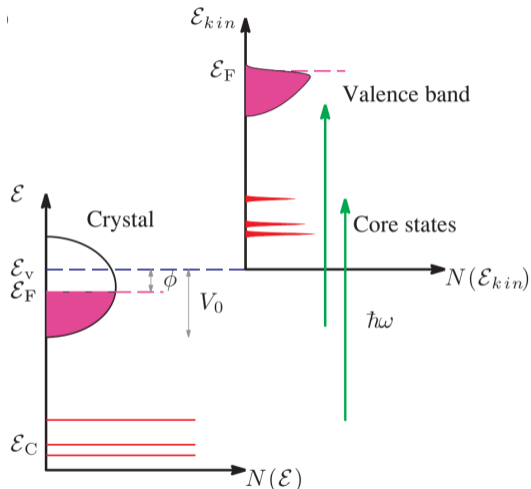
The work function, ϕ , is the minimum energy required to promote an electron from the top of the valence band at the Fermi energy, \mathcal{E}_F , to the vacuum energy, \mathcal{E}_v

The photoemission process



With the incident photon energy, $\hbar\omega$, held constant, an analyzer is used to measure the kinetic energy, \mathcal{E}_{kin} , of the photoelectrons emitted from the surface of the sample

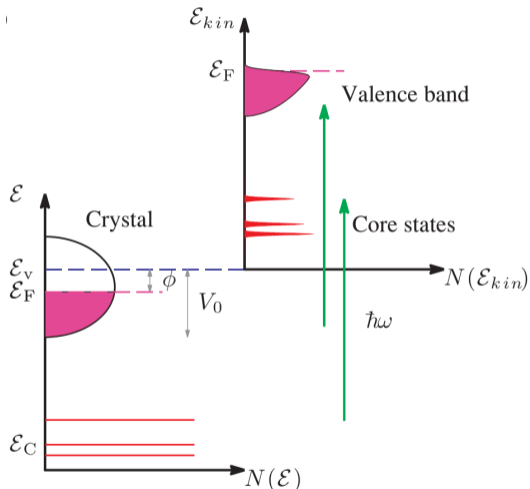
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The photoemission process

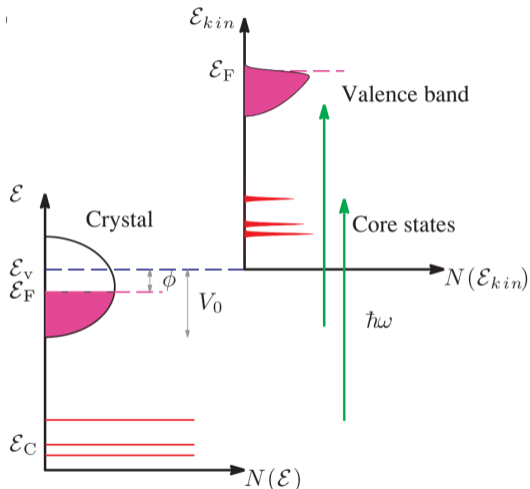


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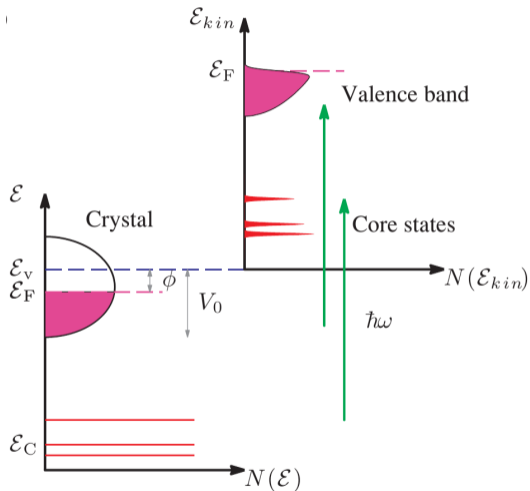
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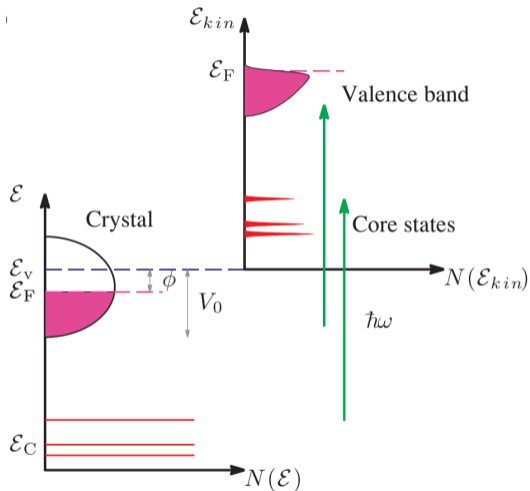
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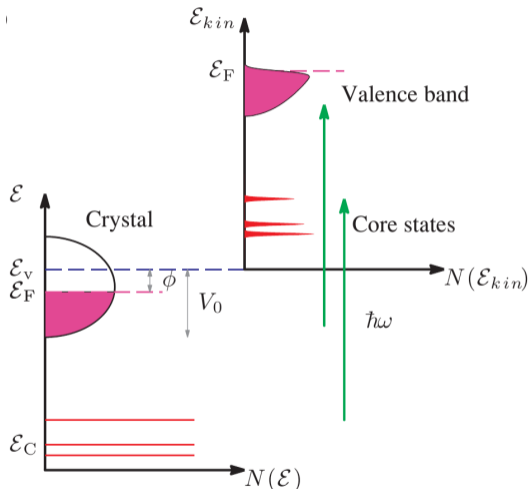
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$$\mathcal{E}_{kin} = \frac{\hbar^2 q_v^2}{2m} = \hbar\omega - \phi - \mathcal{E}_B$$

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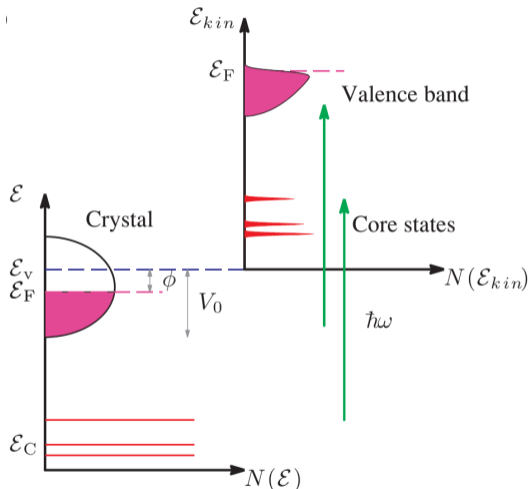
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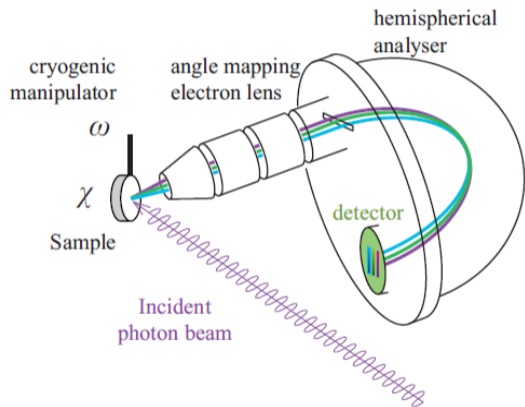
the maximum kinetic energy measured is thus related to the Fermi energy

the core states are used to fingerprint the chemical composition of the sample

Hemispherical mirror analyzer



The electric field between the two hemispheres of radius R_1 and R_2 has a R^2 dependence from the center of the hemispheres



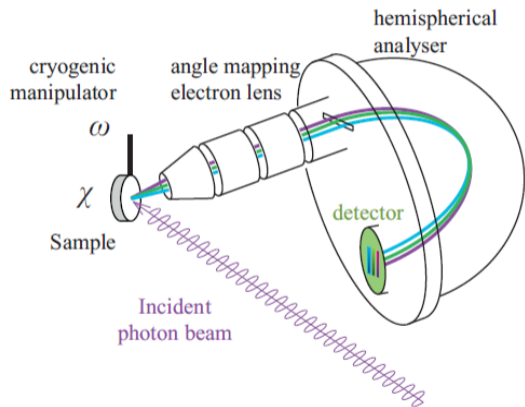
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Electrons with \mathcal{E}_0 , called the “pass energy”, will follow a circular path of radius

$$R_0 = (R_1 + R_2)/2$$



Hemispherical mirror analyzer

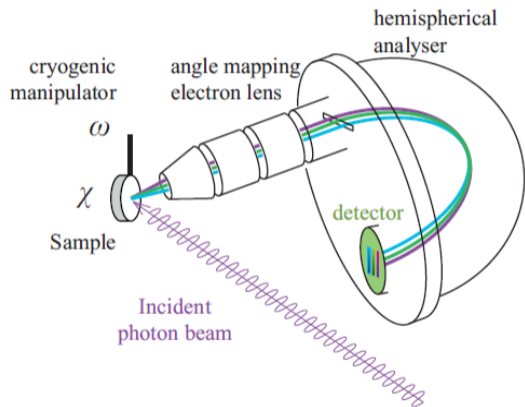


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Electrons with different azimuthal exit angles ω will map to different positions on the 2D detector

